

compressed to 50 kilobars, the relative volume $\Delta V/V \simeq 0.035$ corresponds to a change in lattice parameter of only a few percent.²⁵ The greatest influence on the separation distance would appear to be the increase of G with pressure.

The shock problem, however, is no doubt further complicated by the heating associated with shock propagation,²⁵ since the SFE depends also on the temperature.³⁰ Copley and Kear³¹ also suggest that the separation distance of partial dislocations may change at high rates of strain due to dislocation velocity, and this effect could certainly have a bearing on cross-slip. Thus, it is difficult to assess the relative changes in the SFE as a function of the temperature, pressure, and strain rate in shock loading, and it is not clear whether these effects reinforce or oppose each other.

Summary

A series of α brasses of varying SFE were shock-deformed to determine the influence of SFE in the formation of dislocation configurations in f.c.c. metals. The dislocation configurations observed after moderate shock (55 kilobars) were found not to vary appreciably from those produced by more conventional means of deformation. Apparently the relative amounts of cross-slip were controlled by the SFE in either case. A transition from a planar array of dislocations at low values of SFE to a dislocation cell structure at higher SFE was found to occur in a range 25 to 36 ergs/cm².

References

1. R. D. Heidenreich and W. Shockley, in *Report of Conference on Strength of Solids*, Physical Society, London, 1948, p. 57.
2. M. J. Whelan, *Proc. Roy. Soc.*, *A249* (1959) 114.
3. A. Howie and P. R. Swann, *Phil. Mag.*, *6* (1961) 1215.
4. P. C. J. Gallagher, *J. Appl. Phys.*, *37* (1966) 1710.
5. T. Jossang and J. P. Hirth, *Phil. Mag.*, *13* (1966) 657.
6. V. C. Kannan and G. Thomas, *J. Appl. Phys.*, *37* (1966) 2363.
7. P. R. Swann, in *Electron Microscopy and Strength of Crystals* (G. Thomas and J. Washburn, eds.), Interscience, New York, 1963, p. 131.
8. J. E. Bailey and P. B. Hirsch, *Phil. Mag.*, *5* (1960) 485.
9. S. Mader, in *Electron Microscopy and Strength of Crystals* (G. Thomas and J. Washburn, eds.), Interscience, New York, 1963, p. 183.
10. A. Howie, in *Direct Observations of Imperfections in Crystals* (J. B. Newkirk and J. H. Wernick, eds.), Interscience, New York, 1962, p. 238.
11. G. Thomas, *J. Australian Inst. Metals*, *8* (1963) 80.
12. N. I. Noskova, V. A. Povlov, and S. A. Nemnonov, *Fiz. Metal. i Metalloved.*, *20* (1965) 123.
13. J. W. Edington and R. E. Smallman, *Phil. Mag.*, *11* (1965) 1109.
14. L. M. Brown, *Phil. Mag.*, *10* (1964) 441.
15. B. E. P. Beeston, *Metal Sci.* *2* (1968) 12.
16. E. H. Koster, A. R. Tholen, and A. Howie, *Phil. Mag.* *10* (1964) 1093.

17. M. H. Loretto, L. M. Clarebrough, and R. L. Segall, *Phil. Mag.*, 10 (1964) 731.
18. P. C. J. Gallagher and J. Washburn, *Phil. Mag.*, 14 (1966) 971.
19. R. L. Nolder and G. Thomas, *Acta Met.*, 12 (1964) 227.
20. O. Johari and G. Thomas, *Acta Met.*, 12 (1964) 1153.
21. H. M. Otte and J. R. Holland, in *Interactions between Lattice Defects* (R. R. Hasiguti, ed.), Gordon and Breach, New York, 1967, p. 52.
22. M. C. Inman, L. E. Murr, and M. F. Rose, *ASTM Spec. Tech. Publ. No. 396*, 6 (1966).
23. F. I. Grace, M. C. Inman, and L. E. Murr, *Brit. J. Appl. Phys.*, 1 (1968) 1437.
24. M. F. Rose and F. I. Grace, *Brit. J. Appl. Phys.*, 18 (1967) 671.
25. J. W. Walsh, M. H. Rice, R. G. McQueen, and F. L. Yarger, *Phys. Rev.*, 108 (1957) 196.
26. G. R. Fowles, *J. Appl. Phys.*, 32 (1961) 1475.
27. L. M. Barker, in *Proceedings IUTAM Colloquium on the Behavior of Dense Media under High Dynamic Pressures*, Paris, France, 1967.
28. W. T. Read, *Dislocations in Crystals*, McGraw-Hill, New York, 1953.
29. A. Seeger and O. Buck, *Z. Naturforsch.*, 15A (1960) 1056.
30. T. C. Tisone, J. O. Brittain, and M. Meshii, *Phys. Stat. Sol.*, 27 (1968) 185.
31. S. M. Copley and B. H. Kear, *Acta Met.*, 16 (1968) 227.

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